

# B<sub>4</sub>C/NRL flexible films for thermal neutron shielding

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**Abstract** Boron carbide/natural rubber latex (B<sub>4</sub>C/NRL) flexible films were prepared via dip-molding with B<sub>4</sub>C content in the range of 5–55 wt% for thermal neutron (0.0253 eV) shielding. B<sub>4</sub>C was well dispersed in NRL according to microscopic observation. Both the inside and outside surfaces of the film were smooth. For B<sub>4</sub>C/NRL flexible films, the minimum elongation at break was greater than 600%, the minimum tensile strength was greater than 12 MPa, and the hardness was in the range of 35–55 HA, which were suitable for preparing flexible wearable products. The attenuation efficiencies of the B<sub>4</sub>C/NRL flexible films for thermal neutrons were also calculated. The B<sub>4</sub>C/NRL flexible films exhibit good attenuation effect for thermal neutrons.

**Keywords**  $B_4C \cdot \text{Natural rubber latex} \cdot \text{Thermal neutron} \cdot \text{Shield} \cdot \text{Flexible film}$ 

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#### 1 Introduction

Radioactive rays and related radioactive nuclides have been widely employed in industries. Although radioactive rays provide significant benefits to mankind, they are harmful to human health if appropriate shielding is not employed [1, 2]. Neutrons have a wide range of applications [3, 4]; however, neutron shielding is a challenging work owing to its high radiobiological effect, significant dose weighting factors, and strong penetration abilities [5]. Appropriate protection must be implemented to ensure the health and safety of handlers. A series of materials were investigated to shield against neutrons of different energies [6–8].

In order to avoid the radiation harm caused by thermal neutrons, low Z elements have been commonly used for shielding against thermal neutrons [9]. For example, the investigation by Iskender Akkurt et al. [6] suggested that the boronizing process improved the radiation shielding properties of austenitic stainless steel. The thermal neutron absorption cross-section of  $B_4C$  is 760 barn (1 barn =  $10^{-24}$  cm<sup>2</sup>) [10], which is suitable for shielding against thermal neutrons. Moreover, traditional structural materials containing  $B_4C$  such as metals [11, 12], ceramics [13, 14], concrete [15–17], and polymers [18] have been widely used in neutron shielding owing to their mechanical and chemical stability, abundance, and non-toxicity.

A flexible material in the form of protective shades, protective clothing, protective gloves, and protective helmets is required to shield against neutrons. However, traditional structural materials such as metals, ceramics, concrete, and polymer sheets cannot satisfy these flexible requirements. Therefore, many attempts have been made to develop flexible materials. For example, Hao et al. [19]



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prepared a flexible, flame-retardant composite using a highfunctional methyl vinyl silicone rubber matrix with B<sub>4</sub>C as the neutron absorber. Mersin University prepared ethylene propylene diene monomer (EPDM) rubber with boric acid for neutron shielding [20, 21]. Gwaily et al. [22, 23] prepared B<sub>4</sub>C/NR composites as thermal neutron radiation shields. Ninvong et al. [24] investigated natural rubber (NR) with the addition of boron oxide and boric acid for potential use as a flexible shielding material. In their studies, the preparation of raw materials and methods was aimed at obtaining solid rubber. There are significant differences between solid rubber and rubber latex, such as the preparation method, properties of the product, and applications. Rubber latex, especially natural rubber latex (NRL), is widely used in daily life as a traditional flexible material and can be molded into an arbitrary shape with good plasticity and elasticity. Few works have been reported on NRL as a matrix to shield against thermal neutrons.

In this study, NRL and  $B_4C$  were selected as the matrix and attenuation material, respectively.  $B_4C$  was pre-treated using ball milling to enhance its dispersity and stability in NRL.  $B_4C/NRL$  flexible films with  $B_4C$  content in the range of 5–55 wt% were prepared via dip-molding and were deemed suitable for wearable products. The mechanical properties and thermal neutron shielding properties of the  $B_4C/NRL$  flexible films were analyzed.

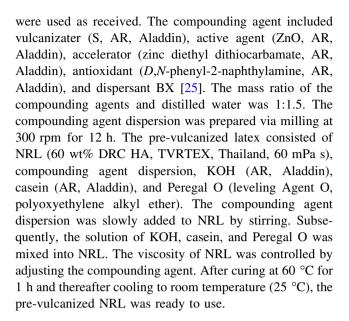
#### 2 Experimental

#### 2.1 Preparation of B<sub>4</sub>C dispersion

The purity of  $B_4C$  (Mudanjiang Qianjin Boron Carbide Co., Ltd) was greater than 99.99%. Commercially available  $B_4C$  powders (3.6  $\mu$ m, 2.45 g cm<sup>-3</sup>) were pre-treated using ball milling at 300 rpm for 12 h. The mass ratio of the corundum ball and material was 2:1. The mass ratio of different diameters of corundum balls was  $\phi$ 20: $\phi$ 10: $\phi$ 6 = 1:3:6. All the other chemicals and analytical reagents were used as received. The  $B_4C$  dispersion consisted of  $B_4C$ , distilled water, ammonia (analytical reagent (AR), Aladdin), and dispersant (two nekal, BX, Aladdin) with a mass ratio of 1:1:0.5:0.01. Ammonia was used to adjust the pH value to enhance the stabilization of the  $B_4C$  dispersion. The dispersant BX was used to increase the wettability of the  $B_4C$  particles and to prevent the re-agglomeration and settlement of  $B_4C$  particles.

#### 2.2 Preparation of pre-vulcanization NRL

The pre-vulcanization composition of the NRL is provided in Table 1. All the chemicals and analytical reagents



#### 2.3 Preparation of B<sub>4</sub>C/NRL flexible films

The  $B_4C$  dispersion was added and dispersed into the pre-vulcanized NRL by stirring. After 48 h, filtration and defoaming were performed to obtain the desired latex (5–55 wt%  $B_4C/NRL$ ). The proper viscosity of the  $B_4C/NRL$  mixture used for dip-molding was 60–100 mPa s.

The B<sub>4</sub>C/NRL flexible films were prepared using dipmolding. Ceramic plates or molds were dipped in NRL for several seconds and thereafter allowed to set. After drying, the molds were vulcanized by boiling water and dried in an oven. Subsequently, the B<sub>4</sub>C/NRL flexible films were ready for thermal neutron shielding.

The thickness of the film can be controlled by dipping multiple times. After a single dip, the thickness of the film ranged from 0.1 to 0.35 mm. The molds were dipped multiple times to achieve the desired thicker films.

### 2.4 Properties characterization

The viscosity of the latex was measured using a digital viscometer (NDJ-5S, Shanghai Ping Xuan Scientific Instrument Co., Ltd) at room temperature. The median diameter (D<sub>50</sub>) of B<sub>4</sub>C was detected using a laser particle size analyzer (Mastersizer 2000) at a scanning speed of 1000 times/s with deionized water as the dispersing agent. The dispersion state of B<sub>4</sub>C in NRL was analyzed using a scanning electron microscope (SEM, JSM6390 LV, Japan JEOL) and an energy dispersive spectrometer (EDS). Fourier transform infrared (FT-IR) spectra were recorded using Nicolet FT-IR Nexus, in the range of 4000–500 cm<sup>-1</sup>. The test mode was set to total reflection. Wide-angle X-ray diffraction (WAXD) was performed on a D/Max-RB target X-ray diffraction (Japan Rigaku) under



Table 1 Dry weight composition of pre-vulcanized NRL

Raw material	NRL	S	ZnO	ZDC	D	Casein	Peregal O	КОН	BX
Content (wt%)	100	1	1	1	0.2	0.1-0.5	0.1-0.5	0.1-0.2	0.1

the following condition: Cu K $\alpha$  radiation ( $\lambda = 0.15406$  nm) at the voltage of 40 kV. The scanning rate was 15°/min in the angle range of 10°–80°. Mechanical properties of the films including the tensile strength and elongation at break were tested using a universal testing machine (CMT6103, Meitesi Industry System (China) Co. Ltd). The harnesses of the films were measured using a shore durometer (LX-A, China Jun Ping Machinery Factory).

# 2.5 Thermal neutron shielding property measurement

The thermal neutron shielding property of the films was evaluated using a 49-2 swimming-pool-type reactor at the Reactor Engineering Research and Design Institute of the Chinese Atomic Energy Academy. The power of the 49-2 reactor was 3500 MW, with light water used as the coolant and moderator. Figure 1 illustrates a schematic diagram of the thermal neutron shielding assessment.

During the measurement, the sample was placed 1600 mm from the thermal column. Three Dy–Al alloy foil detectors were placed on both sides of each sample. The positions of the six foil detectors on both sides of the sample did not overlap. A cadmium foil was used for shielding against lateral neutrons of the thermal column. The diameter of Dy–Al alloy foil was 4 mm with a thickness of 0.3 mm. The performances of Dy–Al alloy foil detectors were as follows: The correction coefficient between the foil detectors was less than 1%, and the Dy detector was only sensitive to thermal neutrons.

Cadmium was used to absorb thermal neutrons, and a small collimator aperture was used to reduce the counting rate. A 3-cm lead brick was used to screen the  $\gamma$ -rays to reduce their impact. The activation cross-section of the neutrons above the thermal neutrons was very small and

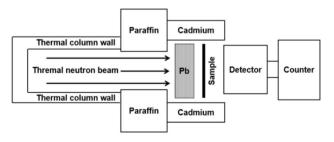


Fig. 1 Schematic diagram of the thermal neutron shielding assessment

could be ignored relative to the thermal neutron activation cross-section. The most probable energy of the thermal neutron was 0.0253 eV. The activity of the foil detector after activation by the neutrons was proportional to the neutron flux rate on the detector. The normalization activity differences between the front and back Dy detectors reflected the thermal neutron shielding performance of the sample. The radioactive decay equation is as follows [16, 17]:

$$A_0 = A \cdot e^{\frac{\ln 2}{T_{1/2}} \cdot t},\tag{1}$$

where  $A_0$  is the initial normalization activity, A is the normalization activity at time t, and  $T_{1/2}$  is half-time.

The thermal neutron shielding performance of the sample was calculated using the following equation:

Shielding efficiency = 
$$(A_{\text{front}} - A_{\text{back}})/A_{\text{front}} \times 100\%$$
, (2)

where  $A_{\text{front}}$  and  $A_{\text{back}}$  represent the normalization activity of the front and back Dy detectors at time t, respectively.

After  $A_{\text{front}}$  and  $A_{\text{back}}$  are obtained, the macroscopic cross-section of the actual material was calculated using Lambert–Beer law (3) [7, 26]:

$$A_{\text{back}} = A_{\text{front}} e^{-\sigma \cdot d}, \tag{3}$$

where d is the thickness of the material and  $\sigma$  is the macroscopic cross-section of the material. The term  $\sigma$  was obtained by measuring  $A_{\rm front}$ ,  $A_{\rm back}$ , and d.

#### 3 Results and discussion

#### 3.1 Dispersion state of B<sub>4</sub>C in NRL

Particles in commercial  $B_4C$  powders agglomerate owing to environmental humidity. The  $B_4C$  used was pretreated via ball milling to avoid particle agglomeration and precipitation. The median diameter  $(D_{50})$  of  $B_4C$  varied with the milling time, which was detected using a laser particle size analyzer. As shown in Fig. 2,  $D_{50}$  of the  $B_4C$  particles reduced from 3.6 to 1  $\mu$ m after milling for 12 h. The granule shape of the  $B_4C$  particles did not change after milling. No apparent precipitation was observed after 10 days, indicating that the stability of the pre-vulcanized latex containing  $B_4C$  is suitable for practical applications.

The SEM images of the flexible film with different contents of B<sub>4</sub>C are shown in Fig. 3. The bright spots in the



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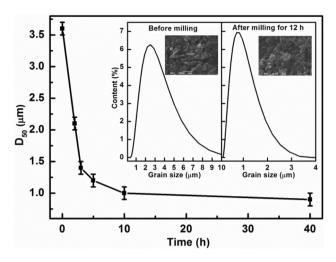


Fig. 2 Median diameter ( $D_{50}$ ) of  $B_4C$  particles with milling time. Inset images: grain size distribution and SEM images of  $B_4C$  particles (a) before milling and (b) after milling for 12 h

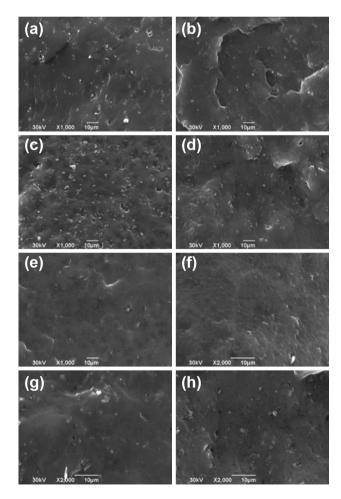


Fig. 3 SEM images of B<sub>4</sub>C/NRL flexible films with **a** 5 wt%, **b** 10 wt%, **c** 15 wt%, **d** 20 wt%, **e** 25 wt%, **f** 35 wt%, **g** 45 wt%, and **h** 55 wt% B<sub>4</sub>C

figure are the  $B_4C$  particles. The small pits are the trails left behind when  $B_4C$  particles peeled off the interface.  $B_4C$  particles are well distributed in NRL with no pores or cracks.

Figure 4 demonstrates the EDS images of the B<sub>4</sub>C/NRL flexible films. The small bright spots in the figure represent the B<sub>4</sub>C particles, indicating that they are well distributed in the NRL. The number of small bright spots increased with the B<sub>4</sub>C concentration. The main component of NRL is a kind of polyisoprene, and its molecular formula is (C<sub>5</sub>H<sub>8</sub>)<sub>n</sub>. The main components of the B<sub>4</sub>C/NRL flexible films are B<sub>4</sub>C and NR. Accordingly, the main elements of the B<sub>4</sub>C/NRL flexible films are B, C, and H. B and C can be detected using EDS; however, H cannot be detected using EDS. Excluding the weight of H, the weights of B and C in B<sub>4</sub>C/NRL obtained using EDS are presented in Table 2. The mass fraction of B in B<sub>4</sub>C is 0.7826. The mass

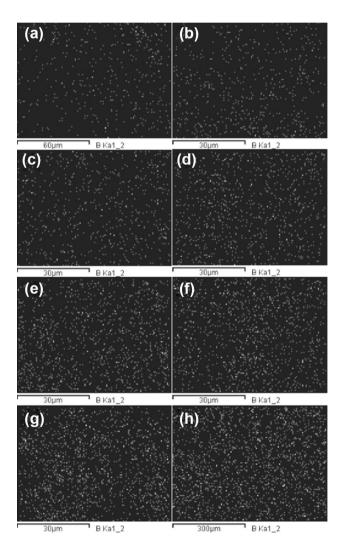


Fig. 4 EDS images of B<sub>4</sub>C/NRL flexible films with a 5 wt%, b 10 wt%, c 15 wt%, d 20 wt%, e 25 wt%, f 35 wt%, g 45 wt%, and h 55 wt% B<sub>4</sub>C



**Table 2** Quantitative analysis data of  $B_4$ C/NRL from EDS

Designed content of B <sub>4</sub> C (wt%)	5	10	15	20	25	35	45	55
B Kα (wt%)	4.19	8.42	12.09	16.39	19.99	27.91	35.26	42.93
C Ka (wt%)	95.81	91.58	87.91	83.61	80.01	72.09	64.74	57.07
Total (wt%)	100	100	100	100	100	100	100	100
Calculated content of B <sub>4</sub> C (wt%)	5.35	10.75	15.45	20.94	25.54	35.66	45.05	54.86
Calculated content of NR (wt%)	94.65	89.25	84.55	79.06	74.46	64.34	54.95	45.14
Total (wt%)	100	100	100	100	100	100	100	100

fraction of  $B_4C$  in  $B_4C/NRL$  is calculated by dividing 0.7826 by the weight of B. In Table 2, the designed contents of  $B_4C$  are consistent with the calculated values.

The SEM images of the outside and inside surfaces of the B<sub>4</sub>C/NRL flexible films are shown in Fig. 5. The inside surface is defined as the film face adjacent to the mold during dipping. The outside surface is defined as the film face away from the mold during dipping. Both the inside and outside surfaces were smooth, and no apparent particles peeled from the matrix. This indicates that the surface of the film is in a good condition and conducive to practical applications.

#### 3.2 WAXD of the B<sub>4</sub>C/NRL films

The WAXD patterns of the  $B_4C/NRL$  flexible films are presented in Fig. 6. The peak observed at  $20^\circ$  revealed the amorphous feature of NR (Fig. 6a). The strongest peak of  $B_4C$  was observed at  $37.6^\circ$  (Fig. 6a). The peak amplitude at  $20^\circ$  became weakened with an increase of  $B_4C$  (Fig. 6b), and thus, the relative peak amplitude (i.e., the ratio of peak amplitudes at  $37.6^\circ/20^\circ$ ) improved (Fig. 6b). The WAXD feature of NRL became weaker with an increase in the concentration of  $B_4C$ , whereas the WAXD feature of  $B_4C$  became stronger. When the content of  $B_4C$  was in the range

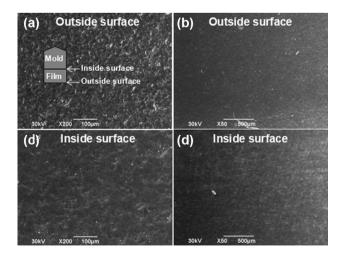
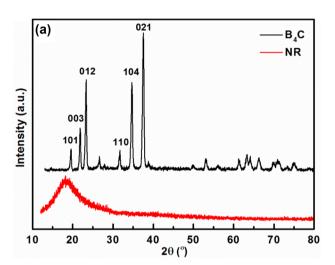


Fig. 5 SEM images of the outside surface a, b and inside surface c, d of a B<sub>4</sub>C/NRL flexible film containing 25 wt% B<sub>4</sub>C



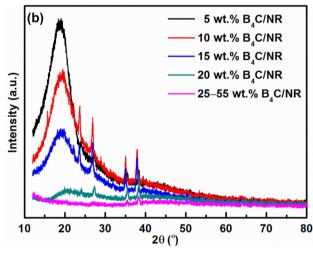


Fig. 6 (Color online) WAXD pattern of  $a\ B_4C$  and NRL, and  $b\ B_4C/$  NRL flexible films

of 25–55 wt%, the WAXD pattern of the  $B_4 C/NRL$  flexible films was weak.

### 3.3 FT-IR of B<sub>4</sub>C/NRL films

The FT-IR spectra of the B<sub>4</sub>C/NRL flexible films are shown in Fig. 7. The addition of B<sub>4</sub>C did not change the peak position of the IR characteristic peak of NRL, indicating that B<sub>4</sub>C did not change the molecular structure of



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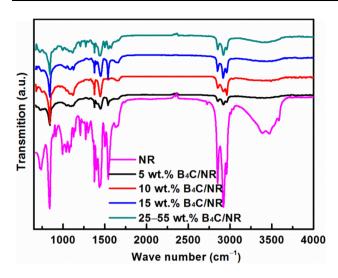


Fig. 7 (Color online) FT-IR spectra of the B<sub>4</sub>C/NRL flexible films

NRL. The wave numbers of the B<sub>4</sub>C/NRL composite are summarized in Table 3. Our results are consistent with the previous reports of IR spectra of NR illustrated as follows: The wave number of nearly 3500 cm<sup>-1</sup> was attributed to the N-H stretching from a small amount of protein in the NR. The protein content decreased after curing in boiling water and subsequent washing. Therefore, the corresponding peak was not apparent in the B<sub>4</sub>C/NRL films. The wave numbers of 2966 and 2913 cm<sup>-1</sup> were attributed to the C-H stretching [27], whereas the wave numbers of 2846, 1665, and 1438 cm<sup>-1</sup> corresponded to the C-H stretching, C=C stretching, and CH<sub>2</sub> deformation, respectively [28]. The wave numbers of 1535 and 1076 cm<sup>-1</sup> were attributed to the O-H bending and C-C stretching, respectively [29]. The wave number of 1380 cm<sup>-1</sup> was attributed to the CH<sub>3</sub> bending [30]. The wave numbers of 1665 and 836 cm<sup>-1</sup> were attributed to the C=C stretching and C=C bending of the isoprene unit, respectively [31].

#### 3.4 Mechanical property of B<sub>4</sub>C/NRL films

 $B_4C/NRL$  flexible films with 5, 10, 15, 20, 25, 35, 45, and 55 wt%  $B_4C$  were vulcanized by boiled water at 100 °C for 40, 60, or 80 min. The mechanical properties are illustrated in Fig. 8. The minimum tensile strength of all the samples was greater than 12 MPa (Fig. 8a), which is higher than that of the reported rubber for thermal neutron shielding [19–23]. The tensile strength of the 10%  $B_4C/NRL$  film was the highest under different vulcanizing temperatures. The tensile strength was the highest when

films with different contents of  $B_4C$  were vulcanized for 60 min. The highest tensile strength among all the samples was 29 MPa. When considering the resulting strength and energy consumption, curing for 60 min was considered the optimal timeframe. As the  $B_4C$  concentration increased, the tensile strength of the  $B_4C/NRL$  films initially increased and thereafter decreased. The tensile strengths of the 5, 10, and 15 wt%  $B_4C/NRL$  films were greater than that of the NRL film. The tensile strengths of the 20, 25, 35, 45, and 55 wt%  $B_4C/NRL$  films were less than that of the NRL film. No other fillers were added to NRL as reinforcing agents. The added  $B_4C$  played the role of a reinforcing agent in the NRL. The addition of 10 wt%  $B_4C$  achieved the best reinforcing effect.

The elongation at break of the  $B_4C/NRL$  films did not vary significantly in the range of 600–810% (Fig. 8b), which is higher than that of the reported rubber for thermal neutron shielding [19–23], indicating that the  $B_4C/NRL$  films are suitable for the high elasticity requirement in some cases, such as gloves.

Figure 9 shows the hardness of the  $B_4C/NRL$  films. The shore hardness of the  $B_4C/NRL$  films increased with the increase in concentration of  $B_4C$ ; however, the  $B_4C/NRL$  films were still flexible for practical applications in comparison with common latex gloves. The hardness is lower than that of the reported rubber for thermal neutron shielding [24].

# 3.5 Thermal neutron shielding property of B<sub>4</sub>C/NRL films

The density of the B<sub>4</sub>C/NRL film was calculated using the following equation:

$$\rho = \frac{m_1 + m_2}{v_1 + v_2} = \frac{\rho_2/x_2}{\rho_2 \cdot x_1/(\rho_1 \cdot x_2) + 1},\tag{4}$$

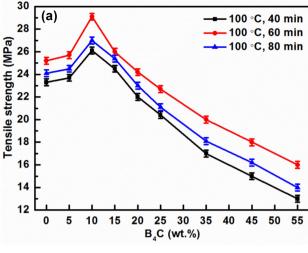
where m, v,  $\rho$ , and x represent the mass, volume, density, and mass fraction, respectively, and subscripts 1 and 2 represent B<sub>4</sub>C and NRL, respectively. Thus,  $m_1/m_2 = x_1/x_2 = \rho_1 v_1/\rho_2 v_2$ , and  $x_1 + x_2 = 1$ . The density of NRL measured using the Archimedes principle was 1 g cm<sup>-3</sup>. The density of the B<sub>4</sub>C/NRL flexible film is shown in Fig. 10. The experimental values are consistent with the calculated values.

Analogous with the attenuation for  $\gamma$ -rays, the mass attenuation coefficient of the B<sub>4</sub>C/NRL composite was calculated using the following equation [7, 32]:

Table 3 Wave numbers of the peaks in the FT-IR spectra of the B<sub>4</sub>C/NRL films

ν (cm <sup>-1</sup> )	2966	2913	2846	1665	1535	1438	1380	1076	839
Vibration mode	ν (C–H)	ν (C–H)	ν (C–H)	v (C=C)	δ (Ο–Η)	δ (CH <sub>2</sub> )	δ (CH <sub>3</sub> )	v (C–C)	δ (C=C)





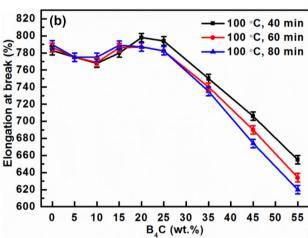


Fig. 8 (Color online) Tensile strength  $\bf a$  and elongation at break  $\bf b$  of the B<sub>4</sub>C/NRL flexible films

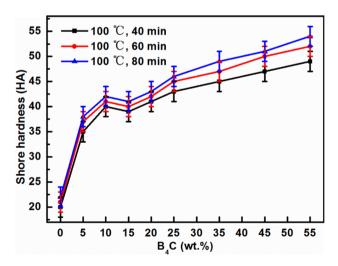
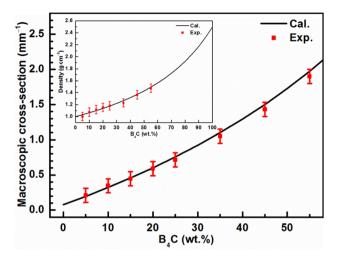


Fig. 9 (Color online) Shore hardness of the B<sub>4</sub>C/NRL flexible films



**Fig. 10** (Color online) Macroscopic cross-section of the B<sub>4</sub>C/NRL flexible films for attenuating thermal neutrons. Inset image: Density of the B<sub>4</sub>C/NRL flexible films

$$\frac{\sigma}{\rho} = \sigma_{\rm m} = \sigma_{\rm m,1} \cdot x_1 + \sigma_{\rm m,1} \cdot x_2,\tag{5}$$

where  $\sigma$  is the line attenuation coefficient or macroscopic cross-section of the composite;  $\sigma_m$  is the mass attenuation coefficient of the  $B_4C/NRL$  composite; the terms  $\sigma_{m,1}$  and  $\sigma_{m,2}$  represent the mass attenuation coefficients of  $B_4C$  and NR, respectively. The mass attenuation coefficient of  $B_4C$  for attenuating thermal neutrons can be obtained from Monte Carlo simulations or the literature database. The macroscopic cross-section of  $B_4C$  for thermal neutrons (0.0253 eV) fitted by an experiment was  $58.537 \text{ cm}^{-1}$  [33]. As NR and high-density polyethylene (HDPE) are both mainly composed of carbon and hydrogen, the macroscopic cross-section of NR for attenuating thermal neutrons (0.0253 eV) was approximately equal to that of HDPE, which was determined by an experiment as  $0.77 \text{ cm}^{-1}$  [34].

The macroscopic cross-section of the  $B_4C/NRL$  composite was calculated using the following equation:

$$\sigma = (\sigma_{m,1} \cdot x_1 + \sigma_{m,2} \cdot x_2) \cdot \frac{\rho_2 / x_2}{\rho_2 \cdot x_1 / (\rho_1 \cdot x_2) + 1}.$$
 (6)

The macroscopic cross-section of the  $B_4C/NRL$  flexible films for attenuating thermal neutrons is shown in Fig. 10. The experimental values of  $\sigma$  were obtained from Eq. (3). The experimental values and the values calculated using Eq. (6) are consistent with each other. However, our results are different from those in the literature [35]. This was mainly because our calculations of the macroscopic cross-section were based on experiments with thermal neutrons of energy 0.0253 eV. The most probable energy of thermal neutrons in the literature was larger than 0.0253 eV. [35].



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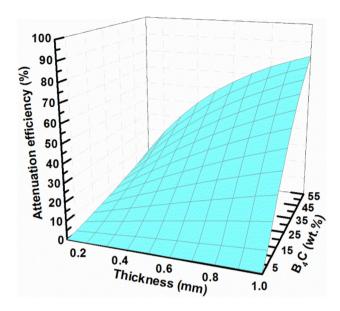


Fig. 11 (Color online) Shielding efficiency of the  $B_4 C/NRL$  flexible films for attenuating thermal neutrons

After the macroscopic cross-section of the  $B_4C/NRL$  flexible films was obtained, the attenuation efficiency of the  $B_4C/NRL$  flexible films for thermal neutrons was calculated for different thicknesses and percentages of  $B_4C$  using Eq. (3). Figure 11 illustrates the attenuation efficiency of the  $B_4C/NRL$  flexible films for attenuating thermal neutrons (0.0253 eV). This result is favorable to engineering design. If the attenuation efficiency is set, suitable thickness and  $B_4C$  content can be designed to satisfy the attenuation efficiency. For a  $B_4C/NRL$  flexible film, the  $B_4C$  content can be calculated by measuring its density and thickness, which is easy to measure; subsequently, the efficiency for attenuating thermal neutrons (0.0253 eV) can be obtained.

## 4 Conclusion

 $B_4C/NRL$  flexible films for thermal neutron shielding were successfully prepared via dip-molding with  $B_4C$  content in the range of 5–55 wt%. The results indicate that  $B_4C$  was well dispersed in NRL. The stability of pre-vulcanized latex containing  $B_4C$  was satisfactory for practical applications. Both the inside and outside surfaces of the film were smooth.

The minimum elongation at break of the B<sub>4</sub>C/NRL flexible films was greater than 600%; the minimum tensile strength of the B<sub>4</sub>C/NRL flexible films was greater than 12 MPa; the hardness of the B<sub>4</sub>C/NRL flexible films was in the range of 35–55 HA. These results indicate that the B<sub>4</sub>C/NRL flexible films are flexible for practical applications.

The experimental and calculated values of the macroscopic cross-section of the B<sub>4</sub>C/NRL flexible films were consistent with each other. The attenuation efficiency of the B<sub>4</sub>C/NRL flexible films for thermal neutrons was calculated for different thicknesses and percentages of B<sub>4</sub>C. The B<sub>4</sub>C/NRL flexible films exhibited good attenuation effect for thermal neutrons. This result is favorable to engineering design. If the attenuation efficiency is set, suitable thickness and B<sub>4</sub>C content can be designed to satisfy the attenuation efficiency.

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